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**CLAIMS**

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**[Claim(s)]**

[Claim 1] silicon substrate top the compound semiconductor substrate made [ the group-III-V-semiconductor layer ] to carry out epitaxial growth -- setting -- the above it consists of the same configuration element as a group-III-V-semiconductor layer -- compared with an III group atom, V group atom is superfluously included in 1.5% or less of range 0.2% or more Compound semiconductor substrate characterized by including at least one or more layers of group-III-V-semiconductor layers.

[Claim 2] silicon substrate top a group-III-V-semiconductor layer is set to the manufacture approach of the compound semiconductor substrate which carries out epitaxial growth -- V group atom is made to incorporate superfluously in 1.5% or less of range 0.2% or more compared with an III group atom, and V group atom is included superfluously The manufacture approach of the compound semiconductor substrate characterized by including the process into which a group-III-V-semiconductor layer is grown up.

[Claim 3] Compared with an III group atom, V group atom is included superfluously. After carrying out epitaxial growth of the group-III-V-semiconductor layer, it is made to grow up after that. The manufacture approach of the compound semiconductor substrate according to claim 2 characterized by giving the heat treatment process in temperature higher than the growth temperature of a group-III-V-semiconductor layer.

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**DETAILED DESCRIPTION**

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**[Detailed Description of the Invention]****[0001]**

[Industrial Application] This invention relates to the compound semiconductor substrate used for light or a high-speed electron device by the detail, and its manufacture approach more about a compound semiconductor substrate and its manufacture approach.

**[0002]**

[Description of the Prior Art] In recent years, epitaxial growth of the compound semiconductor of a different kind is carried out to this on a substrate, and the compound semiconductor substrate which can utilize each advantage in said substrate and compound semiconductor is studied. For example, by carrying out epitaxial growth of the GaAs on a silicon (it being hereafter described as Si) substrate, the compound semiconductor substrate which combines the mechanical strength which Si holds, and the high-speed responsibility which GaAs holds is produced, and the compound semiconductor substrate obtained by such approach is applied to the electron device, the optical device, etc. However, it is easy to spread said rearrangement with the stress which there was a big difference in the lattice constant and coefficient of thermal expansion in Si and GaAs, it originated in this lattice constant difference, and the rearrangement was [ stress ] easy to be formed in the interface of Si substrate and a GaAs layer, and was generated based on said coefficient-of-thermal-expansion difference toward the front face of said GaAs layer. Consequently, many said rearrangements existed in said GaAs layer front face, the crystallinity near [ in said GaAs layer ] the front face fell to it, and there was a problem that it was difficult to demonstrate the function as a device etc. good.

[0003] To insert the rearrangement reduction layer for controlling propagation of a rearrangement into said GaAs layer as an approach of reducing dislocation density [ / near the GaAs layer front face ] is tried. The distorted superlattice layer which has a different presentation from a GaAs layer in this rearrangement reduction layer, the layer by which the impurity was added are examined (application physics: 61 No. 2 and pp(s) 126-133, and 1992).

[0004] The compound with which a lattice constant differs from GaAs layers, such as for example, an InGaAs layer and a GaAsP layer, is used for said distorted superlattice layer, if this distortion superlatticed layer is inserted, it will be distorted, the propagation of said rearrangement will be bent, and reduction of dislocation density [ / near / said / the GaAs layer front face ] will be achieved by [ which arose from now on ] suppressing propagation of this rearrangement. Moreover, if impurities, such as Zn, In, and Si atom, are added in a GaAs layer, propagation of the rearrangement near [ said ] the GaAs layer front face will be controlled by the effectiveness that a crystal becomes hard, and the effectiveness that pinning of said rearrangement is carried out, and reduction of dislocation density will be achieved according to them.

**[0005]**

[Problem(s) to be Solved by the Invention] However, in the compound semiconductor substrate with which the above-mentioned distorted superlattice layer was inserted, the band structure of a GaAs layer tended to change with said distorted superlattice layers, and the technical problem that change of this

band structure tends to have a bad influence on the electrical characteristics near the GaAs layer front face occurred.

[0006] Moreover, in the compound semiconductor substrate with which said impurity was added, after said impurity was spread near the GaAs layer front face from said rearrangement reduction layer, impurities, such as Zn, In, or Si atom, acted as a dopant, and the technical problem that change arose to carrier concentration [ / near the GaAs layer front face ], and it was easy to have a bad influence on electrical characteristics occurred.

[0007] It is made in view of such a technical problem, and this invention is on Si substrate. There is little dislocation density [ / near / the group-III-V-semiconductor layer was made to carry out epitaxial growth / the compound semiconductor substrate front face ], and it also aims change of band structure or carrier concentration at offering the compound semiconductor substrate and its manufacture approach of little high quality excellent in electrical characteristics.

[0008]

[Means for Solving the Problem] the compound semiconductor substrate applied to this invention in order to attain the above-mentioned purpose -- silicon substrate top the compound semiconductor substrate made [ the group-III-V-semiconductor layer ] to carry out epitaxial growth -- setting -- the above it consists of the same configuration element as a group-III-V-semiconductor layer -- compared with an III group atom, V group atom is superfluously included in 1.5% or less of range 0.2% or more (1) characterized by including at least one or more layers of group-III-V-semiconductor layers.

[0009] moreover, the manufacture approach of the compound semiconductor substrate concerning this invention -- silicon substrate top a group-III-V-semiconductor layer is set to the manufacture approach of the compound semiconductor substrate which carries out epitaxial growth -- V group atom is made to incorporate superfluously in 1.5% or less of range 0.2% or more compared with an III group atom, and V group atom is included superfluously (2) characterized by including the process into which a group-III-V-semiconductor layer is grown up.

[0010] Moreover, the manufacture approach of the compound semiconductor substrate concerning this invention contains V group atom superfluously by the manufacture approach of the compound semiconductor substrate the above-mentioned (2) publication compared with an III group atom. After carrying out epitaxial growth of the group-III-V-semiconductor layer, it is made to grow up after that. It is characterized by giving the heat treatment process in temperature higher than the growth temperature of a group-III-V-semiconductor layer (3).

[0011]

[Function] For example, in the case of the GaAs (III-V group compound) layer, it is known by making it grow up at low-temperature (200-300 degrees C) predetermined temperature comparatively using a molecular-beam-epitaxy method that epitaxial growth of the GaAs layer which contains As (III group atom) of superfluous predetermined concentration compared with Ga (V group atom) can be carried out (Thin Solid Films:231 (1993) 61-73). In this case, the superfluous content concentration of As in said GaAs layer is controlled by mainly changing growth temperature.

[0012] It is the typical sectional view which drawing 6 showed in order to explain propagation of the rearrangement in the compound semiconductor substrate containing one layer of GaAs layers which a GaAs layer is made to carry out epitaxial growth on Si substrate, and contain As superfluously in this GaAs layer, and, as for 11 in drawing, Si substrate is shown. On the Si substrate 11, the GaAs layer 12 which carried out epitaxial growth is formed, and the GaAs layer 13 containing superfluous As is formed on the GaAs layer 12. Since the GaAs layer 13 is comparatively grown up at low-temperature predetermined temperature using the above-mentioned molecular-beam-epitaxy method and balking of As from GaAs layer 13 front face is controlled in the case of growth, superfluous As is incorporated by stoichiometric in the GaAs layer 13. Moreover, on the GaAs layer 13, the GaAs layer 14 which carried out epitaxial growth to the GaAs layer 12 like abbreviation is formed, a part of As superfluously incorporated in the GaAs layer 13 in the epitaxial growth process of the GaAs layer 14 will condense, and it will deposit as metal As13a.

[0013] Thus, in the constituted compound semiconductor substrate, pinning of the rearrangement 15

generated in the interface of the Si substrate 11 and the GaAs layer 12 will be carried out by metal As13a. Therefore, except for a part of rearrangement 15a, almost all the rearrangements 15 will be blocked by the GaAs layer 13, and the propagation to the GaAs layer 14 will be prevented, consequently the consistency of rearrangement 15a [ / near the GaAs layer 14 front face ] decreases. While, as for the superfluous content of As in the GaAs layer 13 for decreasing the consistency of rearrangement 15a, 0.2% or more is needed, a upper limit will be restrained by the grown method and 1.5% becomes a upper limit in molecular beam epitaxy.

[0014] According to the compound semiconductor substrate (1) of the above-mentioned configuration, it describes above. Compared with an III group atom, said V group atom is superfluously included in 1.5% or less of range 0.2% or more. Said some of V group atoms may be made to condense and deposit in a group-III-V-semiconductor layer. By this condensation and V group atom particle which deposited The above which carried out epitaxial growth on Si substrate and this Si substrate Pinning of the rearrangement produced in the interface with a group-III-V-semiconductor layer can be carried out. For this reason, said V group atom is included superfluously. It describes above by the group-III-V-semiconductor layer. Propagation of the rearrangement to a group-III-V-semiconductor layer front face can be blocked, consequently it describes above. Dislocation density [ / near the group-III-V-semiconductor layer front face ] can be reduced. Moreover, said V group atom is included superfluously. Since any atoms other than the configuration element of a compound semiconductor layer are not included in a group-III-V-semiconductor layer, it describes above. While being able to lessen effect affect the band structure of a group-III-V-semiconductor layer, the compound semiconductor substrate of the high quality which can also lessen change of carrier concentration, therefore was excellent in electrical characteristics will be obtained.

[0015] Moreover, the process which makes V group atom in the manufacture approach (2) of the compound semiconductor substrate of the above-mentioned configuration incorporate superfluously is easily carried out by adopting a molecular-beam-epitaxy method, and the excessive amount of V group atom may be easily controlled by setup of growth temperature. The above-mentioned compound semiconductor substrate (1) will be easily manufactured by enforcing the manufacture approach (2) of the compound semiconductor substrate of the above-mentioned configuration.

[0016] Moreover, by the manufacture approach (3) of the above-mentioned compound semiconductor substrate, said V group atom is included superfluously. A still larger sludge than that of said V group atom can be formed into a group-III-V-semiconductor layer, consequently propagation of a rearrangement is prevented further, and it describes above. Dislocation density [ / near the group-III-V-semiconductor layer front face ] can manufacture the compound semiconductor substrate reduced further.

[0017]

[Working Example(s) and Comparative Example(s)] Hereafter, the example of the compound semiconductor substrate concerning this invention and its manufacture approach is explained based on a drawing. Drawing 1 is the sectional view having shown typically the example 1 of the compound semiconductor substrate concerning this invention, and 11 in drawing shows Si substrate. On the Si substrate 11, the GaAs layer 12 whose thickness is about 1 micrometer is formed, and the GaAs layer 13 whose thickness which contains As in an excess about 0.4% is about 0.2 micrometers is formed on the GaAs layer 12. On the GaAs layer 13, the GaAs layer 14 whose thickness is about 2 micrometers is formed, and the compound semiconductor substrate 10 is constituted including these Si substrate 11, the GaAs layers 12 and 14, and the GaAs layer 13 that contains As superfluously.

[0018] When manufacturing the compound semiconductor substrate 10 of such a configuration, it is made to grow up by the molecular-beam-epitaxy method, using Solid-state As and Solid-state Ga as a source of vacuum evaporationo. First, after field bearing washes Si substrate which inclined in the [110] directions 2 degrees by about 1% of fluoric acid (HF), it inserts in a molecular beam epitaxy chamber, heats at about 900 degrees C, and carries out evaporation cleaning of the oxide film on Si substrate 11 front face in a field (001). Next, with a two-step grown method, first, about 20 and growth temperature are set as about 400 degrees C, a growth rate is set as about 0.3 micrometer/h for the molecular-beam

intensity ratio to Ga of As, respectively, and GaAs layer (initial layer) 12a whose thickness is about 100nm is grown up on the Si substrate 11. Next, about 20 and growth temperature are set as about 600 degrees C, a growth rate is set as about 1 micrometer/h for the molecular-beam intensity ratio to Ga of As, respectively, GaAs layer 12b whose thickness is about 900nm is grown up, and the GaAs layer 12 of about 1 micrometer of sum totals is formed. Next, about 20 and growth temperature are set as about 200 degrees C, a growth rate is set as about 1 micrometer/h for the molecular-beam intensity ratio to Ga of As, respectively, and epitaxial growth of the GaAs layer 13 whose thickness which contains As superfluously on the GaAs layer 12 is about 0.2 micrometers is carried out. Next, about 20 and growth temperature are set as about 600 degrees C, a growth rate is set as about 1 micrometer/h for the molecular-beam intensity ratio to Ga of As, respectively, and epitaxial growth of the GaAs layer 14 whose thickness is about 2 micrometers is carried out on the GaAs layer 13.

[0019] As a result of measuring the superfluous amount of As(es) in the GaAs layer 13 of the compound semiconductor substrate 10 manufactured by the above-mentioned approach, it was about 0.4%. In addition, this superfluous amount of As(es) was converted and calculated from the lattice constant of the GaAs layer 13 measured according to the X diffraction based on the relation (Thin Solid Films:231 (1993) 61-73 drawing 3 R> 3) between the superfluous amount of As(es) already calculated, and the elongation of a lattice constant.

[0020] Below, the result of having measured the dislocation density and carrier concentration of a compound semiconductor substrate concerning an example 1 is explained. Dislocation density dipped the sample in the melting potassium hydroxide (KOH) kept at about 350 degrees C for about 30 seconds, and asked for it from the number of pits per [ which appeared in GaAs layer 14 front face ] unit area. Moreover, it asked for carrier concentration by the capacity-amplitude measurement. In addition, that in which the GaAs layer 13 in the compound semiconductor substrate 10 is not formed as an example 1 of a comparison was used. Moreover, as an example 2 of a comparison, that in which the superlattice layer whose thickness is about 0.3 micrometers is inserted was used instead of the GaAs layer 13 in the compound semiconductor substrate 10. This superlattice layer grew up about 20nm In0.1 Ga0.9 As layer and about 10nm GaAs layer by turns by the molecular beam epitaxy, and carried out the laminating of the about ten layers of this, and the In0.1 Ga0.9 As layer grew up the molecular-beam intensity ratio to Ga of As by setting about 18 and growth temperature as about 550 degrees C, and setting a growth rate as about 1.1 micrometer/h. Moreover, as an example 3 of a comparison, that in which Si dope layer whose thickness is about 0.2 micrometers is inserted was used instead of the GaAs layer 13 in the compound semiconductor substrate 10. By the molecular beam epitaxy, this Si dope layer is 3 about 3x10<sup>20</sup> pieces/cm in a GaAs layer. It was made to grow up including Si.

[0021] Drawing 2 is the graph which showed the measurement result of the dislocation density in the GaAs layer front face of the compound semiconductor substrate concerning an example 1 and the examples 1-3 of a comparison. The dislocation density of the compound semiconductor substrate 10 concerning an example 1 is abbreviation 5x10<sup>6</sup> cm<sup>-2</sup>, and although the example 1 of a comparison in which the GaAs layer 13 which contains As superfluously is not formed is started, it is decreasing compared with a case (about 3x10<sup>7</sup> cm<sup>-2</sup>). moreover -- although the dislocation density of the compound semiconductor substrate 10 concerning an example 1 starts the example 2 of a comparison in which the superlattice layer was inserted -- a case (about 4x10<sup>6</sup> cm<sup>-2</sup>) and abbreviation -- it was comparable, and although the example 3 of a comparison was started, it was larger than the case a little.

[0022] Moreover, drawing 3 is the curvilinear Fig. having shown the relation between concentration of electrons and the distance from a compound semiconductor substrate front face, and in the case of the thing concerning an example 1, (a) shows the case, although (b) starts the example 3 of a comparison. Separation between the transistor components which a case shows n mold although an example 1 is started so that clearly from drawing 3, the part of the GaAs layer 13 which contains As superfluously has three or less [ 10<sup>13</sup>cm<sup>-2</sup> ] low concentration of electrons (high resistance), therefore were formed in the GaAs layer 14 can be performed easily. On the other hand, although the example 3 of a comparison was started, as for the case, it turned out that n mold as a result of spreading Si in said Si dope layer even in a surface layer is shown, and it is easy to have a bad influence on the GaAs layer which performs

component formation.

[0023] In the compound semiconductor substrate 10 concerning an example 1, some As atoms can be condensed and deposited in the GaAs layer 13 which contains As atomic number in an excess about 0.4%, and pinning of the rearrangement produced by this condensation and As particle which deposited in the interface with the GaAs layer 12 which carried out epitaxial growth to the Si substrate 11 can be carried out so that clearly from these results. For this reason, by the GaAs layer 13 which contains As atom superfluously, propagation of the rearrangement to GaAs layer 14 front face can be blocked, consequently dislocation density [ / near the GaAs layer 14 front face ] can be reduced. Moreover, since any atoms other than a configuration element are not included in the GaAs layer 13 which contains As atom superfluously, there is also little effect affect the band structure of the GaAs layer 14, and the compound semiconductor substrate of the high quality excellent in electrical characteristics also with little change of carrier concentration can be obtained.

[0024] Moreover, the process which makes As atom incorporate superfluously can be easily carried out by adopting a molecular-beam-epitaxy method, and the excessive amount of As atom can be easily controlled by the manufacture approach of the compound semiconductor substrate 10 concerning an example 1 by setup of growth temperature. Therefore, the compound semiconductor substrate 10 can be easily manufactured by enforcing this manufacture approach.

[0025] Next, the compound semiconductor substrate concerning examples 2-7 has the same configuration as the compound semiconductor substrate 10 and abbreviation concerning the example 1 shown in drawing 1. However, in case epitaxial growth of the GaAs layer 13 which contains As superfluously is carried out, the point that the superfluous amount of As(es) is migrating to about 0.2 to about 1.5% of range is different from the thing concerning an example 1 by changing growth temperature from 180 degrees C of abbreviation to 250 degrees C of abbreviation. Below, the result of having measured the dislocation density of GaAs layer 14 front face in the compound semiconductor substrate concerning examples 2-7 is explained. In addition, as an example of a comparison, what contained As atom in the excess about 0.1% was manufactured collectively.

[0026] In the case of the examples 2-7 which contain As atom superfluously in about 0.2 to about 1.5% of range, dislocation density was low, although dislocation density was as high as abbreviation  $2 \times 10^6$  cm<sup>-2</sup> when the superfluous amount of As(es) was the example of a comparison which is 0.1% so that clearly [ drawing 4 / drawing 4 may be the curvilinear Fig. having shown the relation between dislocation density and the superfluous amount of As(es) and ] from this drawing.

[0027] Next, the compound semiconductor substrate concerning an example 8 has the same configuration as the compound semiconductor substrate 10 and abbreviation concerning the example 1 shown in drawing 1. However, before carrying out epitaxial growth of the GaAs layer 13 and growing up the GaAs layer 14 successively after that, the heat treatment process for about 10 minutes (a temperature rise and the fall time are not included) was given at about 700 degrees C higher than the growth temperature of the GaAs layer 14. Drawing 5 is drawing having shown the heat pattern of the compound semiconductor substrate concerning an example 8. The flux of As was irradiated in order to protect that As evaporates from GaAs layer 13 front face in the case of this heat treatment.

[0028] As a result of measuring the dislocation density in the compound semiconductor substrate front face concerning the example 8 which performed this heat treatment, as compared with  $5 \times 10^6$  cm<sup>-2</sup> of the thing of the example 1 which does not heat-treat, it decreased even to  $2 \times 10^6$  cm<sup>-2</sup>.

[0029] With the compound semiconductor substrate concerning an example 8, while the same effectiveness as the case of an example 1 is acquired, some As atoms can be made to be able to condense further in the GaAs layer 13 which contains As atom superfluously, and a large sludge can be deposited, the propagation prevention effectiveness of a rearrangement can be heightened further, therefore dislocation density [ / near the GaAs layer 14 front face ] can be further reduced, so that clearly from the above-mentioned result. Moreover, this heat treatment temperature could be set up lower than the temperature shown in reference (61 application physics No. 2, pp 126-133, and 1992), and was able to aim at compaction of growth time amount, and saving of heater power.

[0030] In addition, although the above-mentioned example explained the case where the GaAs layer 12

and the GaAs layer 13 which contains As superfluously among 14 were made to carry out epitaxial growth for each In the another example, it is [ layer / a GaP layer, / InGaAs ] another instead of the GaAs layers 12 and 14. While a group-III-V-semiconductor layer is made to carry out epitaxial growth It is [ layer / the GaAs layer 13 which contains As superfluously / the GaP layer which contains V group atoms, such as P and As, superfluously instead of, an InGaAs layer, etc. ] another. A group-III-V-semiconductor layer may be made to carry out epitaxial growth.

[0031] Moreover, although the thing of the above-mentioned example explained the case where the one-layer GaAs layer 13 was made to carry out epitaxial growth for each, in the another example, this may be more than two-layer.

[0032]

[Effect of the Invention] If it is in the compound semiconductor substrate (1) concerning this invention as explained in full detail above it consists of the same configuration element as group-III-V-semiconductor layer -- compared with an III group atom, V group atom is superfluously included in 1.5% or less of range 0.2% or more Since at least one or more layers of group-III-V-semiconductor layers are included Above Compared with an III group atom, said V group atom is superfluously included in 1.5% or less of range 0.2% or more. Said some of V group atoms can be condensed and deposited in a group-III-V-semiconductor layer. By this condensation and V group atom particle which deposited The above which carried out epitaxial growth on Si substrate and this Si substrate Pinning of the rearrangement produced in the interface with a group-III-V-semiconductor layer can be carried out. For this reason, said V group atom is included superfluously. It describes above by the group-III-V-semiconductor layer. Propagation of the rearrangement to a group-III-V-semiconductor layer front face can be blocked, consequently it describes above. Dislocation density [ / near the group-III-V-semiconductor layer front face ] can be reduced. Moreover, said V group atom is included superfluously. Since any atoms other than a configuration element are not included in a group-III-V-semiconductor layer, it describes above. While being able to lessen effect affect the band structure of a group-III-V-semiconductor layer, the compound semiconductor substrate of the high quality which could also lessen change of carrier concentration, therefore was excellent in electrical characteristics can be obtained.

[0033] Moreover, if it is in the manufacture approach (2) of a compound semiconductor substrate V group atom is made to incorporate superfluously in 1.5% or less of range 0.2% or more compared with III group atom, and V group atom is included superfluously. Since at least one or more layers of group-III-V-semiconductor layers are grown up The process which makes said V group atom incorporate superfluously can be easily carried out by adopting the molecule epitaxy method, and the excessive amount of said V group atom can be easily controlled by setup of growth temperature. Therefore, by enforcing this manufacture approach, the above-mentioned compound semiconductor substrate (1) can be manufactured easily.

[0034] Moreover, if it is in the manufacture approach (3) of a compound semiconductor substrate Compared with III group atom, V group atom is included superfluously. After carrying out epitaxial growth of the group-III-V-semiconductor layer, it is made to grow up after that. Since the heat treatment process in temperature higher than the growth temperature of a group-III-V-semiconductor layer is given Said V group atom is included superfluously. A still larger sludge than that of said V group atom can be formed into a group-III-V-semiconductor layer. Consequently, propagation of a rearrangement can be prevented further and described above. Dislocation density [ / near the group-III-V-semiconductor layer front face ] can manufacture the compound semiconductor substrate reduced further.

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**\* NOTICES \***

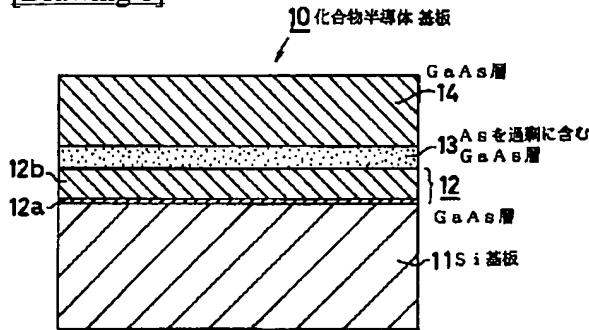
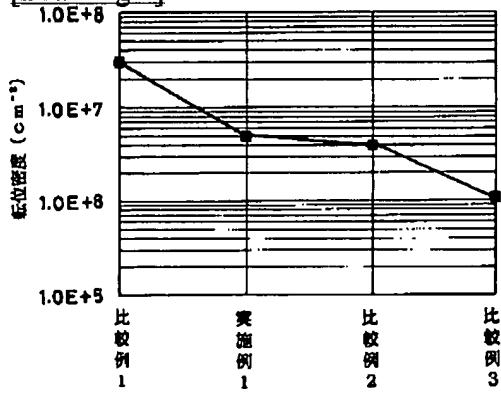
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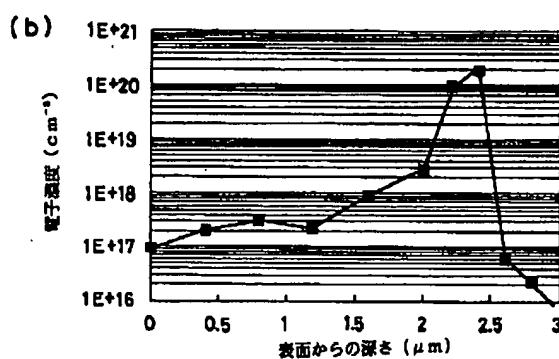
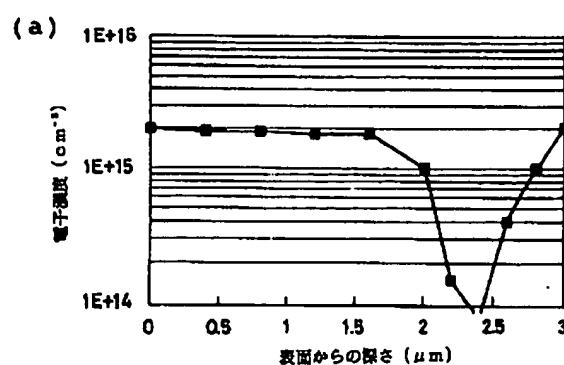
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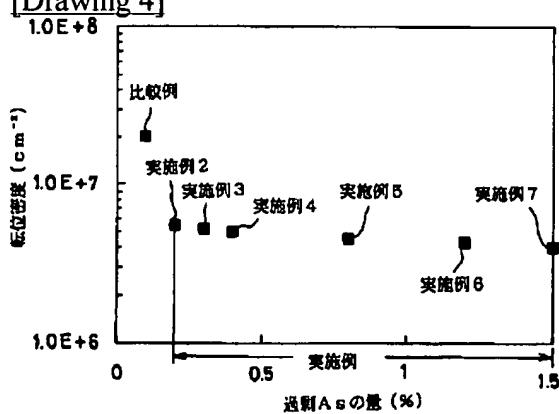
**DRAWINGS**

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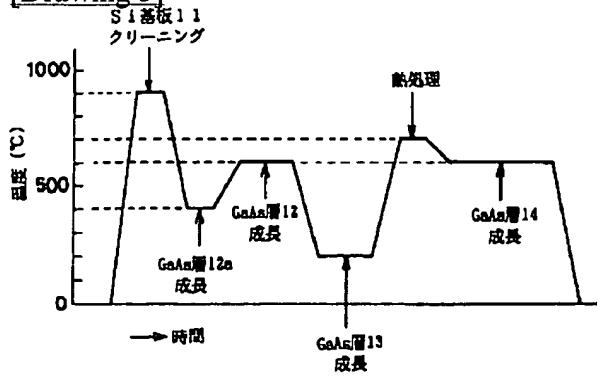
**[Drawing 1]****[Drawing 2]****[Drawing 3]**



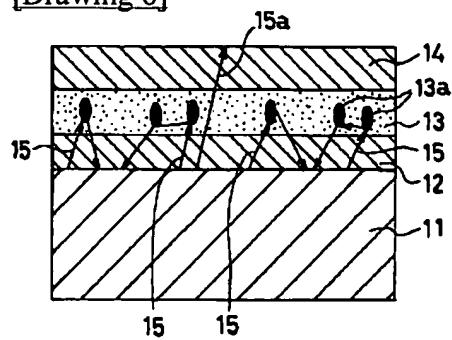
[Drawing 4]



[Drawing 5]



[Drawing 6]



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[Translation done.]

# PATENT ABSTRACTS OF JAPAN

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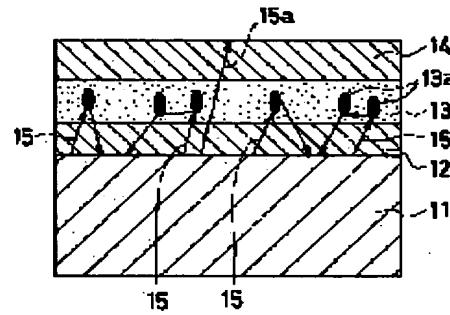
(21)Application number : 06-227836 (71)Applicant : SUMITOMO METAL IND LTD  
 (22)Date of filing : 22.09.1994 (72)Inventor : KATAHAMA HISASHI

## (54) COMPOUND SEMICONDUCTOR SUBSTRATE AND ITS MANUFACTURE

### (57)Abstract:

**PURPOSE:** To lessen the dislocation density and also, lessen the band structure and the change of carrier concentration by providing a III-V compound semiconductor 9 layer which contains group V atoms excessively within the specified range as compared with group III atoms by at least one layer or more.

**CONSTITUTION:** A GaAs layer 12 epitaxially grown is made on an Si substrate 11. And, a GaAs layer 13 containing excessive As is made on the GaAs layer 12. During the growth, since the release of As from the surface of the GaAs layer 13 is suppressed, stoichiometrically excessive As is taken into the GaAs layer 13. The content of excessive As within the GaAs layer 13 for reducing the density of dislocation 15a should be 0.2% or more, while the upper limit value is limited by growth method, so in molecular beam epitaxial growth method, 1.5% becomes the upper limit value.



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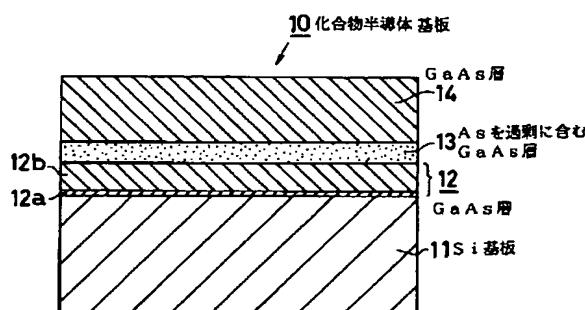
(74)代理人 弁理士 井内 龍二

(54)【発明の名称】 化合物半導体基板及びその製造方法

(57)【要約】

【構成】 GaAs層12、14と同一の構成元素から成り、Ga原子に比べてAs原子を0.2%以上1.5%以下の範囲で過剰に含むGaAs層13を1層含んでいる化合物半導体基板10。

【効果】 GaAs層13中にAs原子の一部が析出させることができ、この析出したAs粒子により、Si基板11とGaAs層12との界面に生じた転位をピンニングさせることができるので、転位の伝播をブロックすることができ、GaAs層14表面近傍における転位密度を低減することができる。またGaAs層13には構成元素以外の原子を含んでいないため、GaAs層14のバンド構造に及ぼす影響も少なく、キャリア濃度の変化も少ない、電気的特性に優れた高品質の化合物半導体基板10を得ることができる。



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## 【特許請求の範囲】

【請求項1】 シリコン基板上に III-V族化合物半導体層がエピタキシャル成長させられた化合物半導体基板において、前記 III-V族化合物半導体層と同一の構成元素から成り、III族原子に比べてV族原子を0.2%以上1.5%以下の範囲で過剰に含む III-V族化合物半導体層を少なくとも1層以上含んでいることを特徴とする化合物半導体基板。

【請求項2】 シリコン基板上に III-V族化合物半導体層をエピタキシャル成長させる化合物半導体基板の製造方法において、III族原子に比べてV族原子を0.2%以上1.5%以下の範囲で過剰に取り込ませ、V族原子を過剰に含む III-V族化合物半導体層を成長させる工程を含むことを特徴とする化合物半導体基板の製造方法。

【請求項3】 III族原子に比べてV族原子を過剰に含む III-V族化合物半導体層をエピタキシャル成長させた後、その後に成長させる III-V族化合物半導体層の成長温度よりも高い温度での熱処理工程を施すことを特徴とする請求項2記載の化合物半導体基板の製造方法。

## 【発明の詳細な説明】

## 【0001】

【産業上の利用分野】 本発明は化合物半導体基板及びその製造方法に関し、より詳細には、例えば光または高速電子デバイス等に使用される化合物半導体基板及びその製造方法に関する。

## 【0002】

【従来の技術】 近年、基板上にこれとは異種の化合物半導体をエピタキシャル成長させ、前記基板及び化合物半導体におけるそれぞれの長所を活用可能な化合物半導体基板が研究されている。例えばシリコン（以下、Siと記す）基板上にGaAsをエピタキシャル成長させることにより、Siが保有する機械的強度と、GaAsが保有する高速応答性とを兼ね備えた化合物半導体基板が作製されており、このような方法によって得られた化合物半導体基板は電子デバイス、光デバイス等に応用されている。しかしSiとGaAsとにおける格子定数及び熱膨張係数には大きな差があり、この格子定数差に起因してSi基板とGaAs層との界面に転位が形成されやすく、また前記熱膨張係数差に基づいて発生した応力により前記転位が前記GaAs層の表面に向かって伝播しやすい。この結果、前記GaAs層表面には前記転位が多く存在し、前記GaAs層における表面近傍の結晶性が低下し、デバイス等としての機能を良好に発揮せしめが難しいという問題があった。

【0003】 GaAs層表面近傍における転位密度を低減する方法として、前記GaAs層中に転位の伝播を抑制するための転位低減層を挿入することが試みられており。この転位低減層にはGaAs層とは異なる組成を有する歪超格子層や、不純物が添加された層等が検討され

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ている（応用物理：61巻2号、pp126-133、1992）。

【0004】 前記歪超格子層には、例えばInGaAs層やGaAsP層等のようなGaAs層とは格子定数が異なる化合物が用いられており、この歪超格子層が挿入されると、これから生じた歪で前記転位の伝播方向が曲げられ、該転位の伝播が抑えられることにより、前記GaAs層表面近傍における転位密度の低減が図られる。またGaAs層中に例えばZn、In、Si原子等の不純物が添加されると、結晶が硬くなる効果と前記転位がピンニングされる効果とにより、前記GaAs層表面近傍への転位の伝播が抑制され、転位密度の低減が図られる。

## 【0005】

【発明が解決しようとする課題】 しかしながら上記した歪超格子層が挿入された化合物半導体基板においては、前記歪超格子層によりGaAs層のバンド構造が変化し易く、このバンド構造の変化がGaAs層表面近傍の電気的特性に悪影響を及ぼし易いという課題があった。

【0006】 また前記不純物が添加された化合物半導体基板においては、前記不純物が前記転位低減層からGaAs層表面近傍に拡散してしまうと、Zn、InまたはSi原子等の不純物がドーパントとして作用し、GaAs層表面近傍におけるキャリア濃度に変化が生じて電気的特性に悪影響を及ぼし易いという課題があった。

【0007】 本発明はこのような課題に鑑みなされたものであり、Si基板上にIII-V族化合物半導体層がエピタキシャル成長させられた化合物半導体基板表面近傍における転位密度が少なく、バンド構造やキャリア濃度の変化も少ない、電気的特性に優れた高品質の化合物半導体基板及びその製造方法を提供することを目的としている。

## 【0008】

【課題を解決するための手段】 上記目的を達成するため本発明に係る化合物半導体基板は、シリコン基板上にIII-V族化合物半導体層がエピタキシャル成長させられた化合物半導体基板において、前記III-V族化合物半導体層と同一の構成元素から成り、III族原子に比べてV族原子を0.2%以上1.5%以下の範囲で過剰に含む III-V族化合物半導体層を少なくとも1層以上含んでいることを特徴としている（1）。

【0009】 また本発明に係る化合物半導体基板の製造方法は、シリコン基板上にIII-V族化合物半導体層をエピタキシャル成長させる化合物半導体基板の製造方法において、III族原子に比べてV族原子を0.2%以上1.5%以下の範囲で過剰に取り込ませ、V族原子を過剰に含む III-V族化合物半導体層を成長させる工程を含むことを特徴としている（2）。

【0010】 また本発明に係る化合物半導体基板の製造方法は、上記（2）記載の化合物半導体基板の製造方法により、III族原子に比べてV族原子を過剰に含む III

-V族化合物半導体層をエピタキシャル成長させた後、その後に成長させる III-V族化合物半導体層の成長温度よりも高い温度での熱処理工程を施すことを特徴としている(3)。

## 【0011】

【作用】例えばGaAs(III-V族化合物)層の場合、分子線エピタキシ法を用いて比較的低温(200~300°C)の所定温度で成長させることにより、Ga(V族原子)に比べて過剰な所定濃度のAs(III族原子)を含むGaAs層をエピタキシャル成長させ得ることが知られている(Thin Solid Films:231(1993)61-73)。この場合、主として成長温度を変えることにより、前記GaAs層中におけるAsの過剰含有濃度を制御している。

【0012】図6はSi基板上にGaAs層がエピタキシャル成長させられ、このGaAs層中にAsを過剰に含むGaAs層を1層含んでいる化合物半導体基板における転位の伝播を説明するために示した模式的断面図であり、図中11はSi基板を示している。Si基板11上にはエピタキシャル成長させられたGaAs層12が形成され、GaAs層12上には過剰のAsを含有するGaAs層13が形成されている。GaAs層13は上記した分子線エピタキシ法を用いて比較的低温の所定温度で成長させられており、成長の際、GaAs層13表面からのAsの離脱が抑制されるため、GaAs層13中には化学量論的に過剰のAsが取り込まれている。またGaAs層13上にはGaAs層12と略同様にエピタキシャル成長させられたGaAs層14が形成されており、GaAs層14のエピタキシャル成長工程において、GaAs層13中に過剰に取り込まれたAsの一部が凝集し、金属As13aとして析出することとなる。

【0013】このように構成された化合物半導体基板では、Si基板11とGaAs層12との界面に発生した転位15が金属As13aによりピンニングされることとなる。したがって一部の転位15aを除き、ほとんどの転位15がGaAs層13によりブロックされてGaAs層14への伝播が阻止されることとなり、この結果、GaAs層14表面近傍における転位15aの密度が減少する。転位15aの密度を減少させるためのGaAs層13中における過剰なAsの含有量は0.2%以上が必要となる一方、上限値は成長法により制約されることとなり、分子線エピタキシャル成長法においては1.5%が上限値となる。

【0014】上記構成の化合物半導体基板(1)によれば、前記III族原子に比べて前記V族原子を0.2%以上1.5%以下の範囲で過剰に含むIII-V族化合物半導体層中に前記V族原子の一部を凝集・析出させ得ることとなり、該凝集・析出したV族原子粒子により、Si基板と該Si基板上にエピタキシャル成長させられた前

記III-V族化合物半導体層との界面に生じた転位をピンニングさせ得ることとなる。このため、前記V族原子を過剰に含むIII-V族化合物半導体層により、前記II-V族化合物半導体層表面への転位の伝播をブロックし得ることとなり、この結果、前記III-V族化合物半導体層表面近傍における転位密度を低減し得ることとなる。また前記V族原子を過剰に含むIII-V族化合物半導体層には化合物半導体層の構成元素以外の原子を含んでいないため、前記III-V族化合物半導体層のバンド構造に及ぼす影響を少なくし得ると共に、キャリア濃度の変化も少なくし得ることとなり、したがって電気的特性に優れた高品質の化合物半導体基板が得られることとなる。

【0015】また上記構成の化合物半導体基板の製造方法(2)におけるV族原子を過剰に取り込ませる工程は、分子線エピタキシ法を採用することにより容易に実施され、またV族原子の過剰量は成長温度の設定により容易に制御され得る。上記構成の化合物半導体基板の製造方法(2)を実施することにより上記化合物半導体基板(1)が容易に製造されることとなる。

【0016】また上記した化合物半導体基板の製造方法(3)により、前記V族原子を過剰に含むIII-V族化合物半導体層中に前記V族原子のより一層大きい析出物を形成し得ることとなり、この結果、転位の伝播がより一層防止され、前記III-V族化合物半導体層表面近傍における転位密度がより一層低減された化合物半導体基板を製造し得ることとなる。

## 【0017】

【実施例及び比較例】以下、本発明に係る化合物半導体基板及びその製造方法の実施例を図面に基づいて説明する。図1は本発明に係る化合物半導体基板の実施例1を模式的に示した断面図であり、図中11はSi基板を示している。Si基板11上には厚さが約1μmのGaAs層12が形成され、GaAs層12上にはAsを約0.4%過剰に含む厚さが約0.2μmのGaAs層13が形成されている。GaAs層13上には厚さが約2μmのGaAs層14が形成されており、これらSi基板11、GaAs層12、14、Asを過剰に含むGaAs層13を含んで化合物半導体基板10が構成されている。

【0018】このような構成の化合物半導体基板10を製造する場合、蒸着源として固体As及び固体Gaを用い、分子線エピタキシ法により成長させる。まず、面方位が(001)面で、[110]方向に2°傾いたSi基板を約1%のフッ酸(HF)で洗浄した後、分子線エピタキシチャンバーに挿入し、約900°Cに加熱してSi基板11表面上の酸化膜を蒸発クリーニングする。次に二段階成長法により、まずAsのGaに対する分子線強度比を約2.0、成長温度を約400°C、成長速度を約0.3μm/hにそれぞれ設定し、Si基板11上に厚

さが約100nmのGaAs層(初期層)12aを成長させる。次にAsのGaに対する分子線強度比を約20、成長温度を約600°C、成長速度を約1μm/hにそれぞれ設定し、厚さが約900nmのGaAs層12bを成長させ、合計約1μmのGaAs層12を形成する。次にAsのGaに対する分子線強度比を約20、成長温度を約200°C、成長速度を約1μm/hにそれぞれ設定し、GaAs層12上にAsを過剰に含む厚さが約0.2μmのGaAs層13をエピタキシャル成長させる。次にAsのGaに対する分子線強度比を約20、成長温度を約600°C、成長速度を約1μm/hにそれぞれ設定し、GaAs層13上に厚さが約2μmのGaAs層14をエピタキシャル成長させる。

【0019】上記の方法により製造された化合物半導体基板10のGaAs層13中における過剰のAs量を測定した結果、約0.4%であった。なおこの過剰のAs量は、すでに求められている過剰なAs量と格子定数の伸びとの関係(Thin Solid Films:231(1993)61-73図3)をもとに、X線回折により測定したGaAs層13の格子定数から換算して求めた。

【0020】以下に、実施例1に係る化合物半導体基板の転位密度及びキャリア濃度を測定した結果について説明する。転位密度は約350°Cに保った溶融水酸化カリウム(KOH)に試料を約30秒間浸し、GaAs層14表面に現れた単位面積当たりのピット数から求めた。またキャリア濃度は容量-電圧測定により求めた。なお比較例1として、化合物半導体基板10におけるGaAs層13が形成されていないものを用いた。また比較例2として、化合物半導体基板10におけるGaAs層13の代わりに、厚さが約0.3μmの超格子層が挿入されているものを用いた。この超格子層は、約20nmのIn<sub>0.1</sub>Ga<sub>0.9</sub>As層と約10nmのGaAs層とを分子線エピタキシャル法により交互に成長させ、これを10層ほど積層したものであり、In<sub>0.1</sub>Ga<sub>0.9</sub>As層はAsのGaに対する分子線強度比を約18、成長温度を約550°C、成長速度を約1.1μm/hに設定することにより成長させた。また比較例3として、化合物半導体基板10におけるGaAs層13の代わりに、厚さが約0.2μmのSiドープ層が挿入されているものを用いた。このSiドープ層は分子線エピタキシャル法により、GaAs層中に約3×10<sup>20</sup>個/cm<sup>3</sup>のSiを含んで成長させた。

【0021】図2は実施例1及び比較例1～3に係る化合物半導体基板のGaAs層表面における転位密度の測定結果を示したグラフである。実施例1に係る化合物半導体基板10の転位密度は約5×10<sup>6</sup>cm<sup>-2</sup>であり、Asを過剰に含むGaAs層13が形成されていない比較例1に係るもの場合(約3×10<sup>7</sup>cm<sup>-2</sup>)に比べて減少している。また実施例1に係る化合物半導体基板10の転位密度は超格子層が挿入された比較例2に係る

ものの場合(約4×10<sup>6</sup>cm<sup>-2</sup>)と略同程度であり、また比較例3に係るものとの場合よりは幾分大きかった。

【0022】また図3は電子濃度と化合物半導体基板表面からの距離との関係を示した曲線図であり、(a)は実施例1に係るものの場合、(b)は比較例3に係るものの場合を示している。図3から明らかなように、実施例1に係るもの場合はn型を示し、Asを過剰に含むGaAs層13の部分は10<sup>13</sup>cm<sup>-3</sup>以下の低い電子濃度(高抵抗)になっており、したがってGaAs層14に形成されたトランジスタ素子間の分離を容易に行える。一方比較例3に係るもの場合は前記Siドープ層中のSiが表面層にまで拡散した結果n型を示し、素子形成を行うGaAs層に悪影響を及ぼし易いことが分かった。

【0023】これらの結果から明らかなように、実施例1に係る化合物半導体基板10では、As原子数を約0.4%過剰に含むGaAs層13中にAs原子の一部を凝集・析出させることができ、この凝集・析出したAs粒子により、Si基板11とエピタキシャル成長させたGaAs層12との界面に生じた転位をピンニングさせることができる。このため、As原子を過剰に含むGaAs層13により、GaAs層14表面への転位の伝播をブロックすることができ、この結果、GaAs層14表面近傍における転位密度を低減することができる。またAs原子を過剰に含むGaAs層13には構成元素以外の原子を含んでいないため、GaAs層14のバンド構造に及ぼす影響も少なく、キャリア濃度の変化も少ない、電気的特性に優れた高品質の化合物半導体基板を得ることができる。

【0024】また実施例1に係る化合物半導体基板10の製造方法では、As原子を過剰に取り込ませる工程は分子線エピタキシ法を採用することにより容易に実施することができ、またAs原子の過剰量は成長温度の設定により容易に制御することができる。したがってこの製造方法を実施することにより化合物半導体基板10を容易に製造することができる。

【0025】次に実施例2～7に係る化合物半導体基板は、図1に示した実施例1に係る化合物半導体基板10と略同様の構成を有している。しかし、Asを過剰に含むGaAs層13をエピタキシャル成長させる際、成長温度を略180°Cから略250°Cまで変化させることにより、過剰のAs量が約0.2%から約1.5%の範囲にわたっている点が実施例1に係るものと相違している。以下に、実施例2～7に係る化合物半導体基板におけるGaAs層14表面の転位密度を測定した結果について説明する。なお比較例として、As原子を約0.1%過剰に含んだものを併せて製造した。

【0026】図4は転位密度と過剰のAs量との関係を示した曲線図であり、この図から明らかなように、過剰なAs量が0.1%の比較例の場合は転位密度が約2×

$10^6 \text{ cm}^{-2}$  と高かったが、As 原子を約 0.2% から約 1.5% の範囲で過剰に含む実施例 2～7 の場合は転位密度が低かった。

【0027】次に実施例 8 に係る化合物半導体基板は、図 1 に示した実施例 1 に係る化合物半導体基板 10 と略同様の構成を有している。しかし、GaAs 層 13 をエピタキシャル成長させ、その後引き続き GaAs 層 14 を成長させる前に、GaAs 層 14 の成長温度よりも高い約 700°C で約 10 分間（温度上昇及び下降時間は含まず）の熱処理工程を施しておいた。図 5 は実施例 8 に係る化合物半導体基板のヒートパターンを示した図である。この熱処理の際、GaAs 層 13 表面から As が蒸発するのを防ぐため、As のフラックスを照射した。

【0028】この熱処理を施した実施例 8 に係る化合物半導体基板表面における転位密度を測定した結果、熱処理を施さない実施例 1 のものの  $5 \times 10^6 \text{ cm}^{-2}$  に比較し、 $2 \times 10^6 \text{ cm}^{-2}$  にまで減少した。

【0029】上記結果から明らかのように、実施例 8 に係る化合物半導体基板では、実施例 1 の場合と同様の効果が得られると共に、As 原子を過剰に含む GaAs 層 13 中に As 原子の一部をより一層凝集させ、大きい析出物を析出させることができ、この結果、転位の伝播防止効果を一層高めることができ、したがって GaAs 層 14 表面近傍における転位密度を一層低減することができる。またこの熱処理温度は文献（応用物理 61巻 2 号、pp126-133、1992）に示されている温度よりも低く設定することができ、成長時間の短縮や、ヒーターパワーの節約を図ることができた。

【0030】なお上記した実施例では、いずれも GaAs 層 12、14 間に As を過剰に含む GaAs 層 13 がエピタキシャル成長させられた場合について説明したが、別の実施例では GaAs 層 12、14 の代わりに GaP 層や InGaAs 層等の別の III-V 族化合物半導体層がエピタキシャル成長させられるとともに、As を過剰に含む GaAs 層 13 の代わりに P や As 等の V 族原子を過剰に含む GaP 層や InGaAs 層等の別の I-II-V 族化合物半導体層がエピタキシャル成長させられたものであってもよい。

【0031】また上記した実施例のものでは、いずれも 1 層の GaAs 層 13 がエピタキシャル成長させられた場合について説明したが、別の実施例ではこれが 2 層以上であってもよい。

### 【0032】

【発明の効果】以上詳述したように本発明に係る化合物半導体基板（1）にあっては、III-V 族化合物半導体層と同一の構成元素から成り、III 族原子に比べて V 族原子を 0.2% 以上 1.5% 以下の範囲で過剰に含む I-II-V 族化合物半導体層を少なくとも 1 層以上含んでいるので、前記 III 族原子に比べて前記 V 族原子を 0.2% 以上 1.5% 以下の範囲で過剰に含む III-V 族化合物

物半導体層中に前記 V 族原子の一部を凝集・析出させることができ、該凝集・析出した V 族原子粒子により、Si 基板と該 Si 基板上にエピタキシャル成長させた前記 III-V 族化合物半導体層との界面に生じた転位をピンニングさせることができる。このため、前記 V 族原子を過剰に含む III-V 族化合物半導体層により、前記 III-V 族化合物半導体層表面への転位の伝播をブロックすることができ、この結果、前記 III-V 族化合物半導体層表面近傍における転位密度を低減することができる。また前記 V 族原子を過剰に含む III-V 族化合物半導体層には構成元素以外の原子を含んでいないため、前記 I-II-V 族化合物半導体層のバンド構造に及ぼす影響を少なくすることができると共に、キャリア濃度の変化も少なくすることができ、したがって電気的特性に優れた高品質の化合物半導体基板を得ることができる。

【0033】また化合物半導体基板の製造方法（2）にあっては、III 族原子に比べて V 族原子を 0.2% 以上 1.5% 以下の範囲で過剰に取り込ませ、V 族原子を過剰に含む III-V 族化合物半導体層を少なくとも 1 層以上成長させるので、前記 V 族原子を過剰に取り込ませる工程は、分子エピタキシ法を採用することにより容易に実施することができ、また前記 V 族原子の過剰量は成長温度の設定により容易に制御することができる。したがってこの製造方法を実施することにより、上記化合物半導体基板（1）を容易に製造することができる。

【0034】また化合物半導体基板の製造方法（3）にあっては、III 族原子に比べて V 族原子を過剰に含む I-II-V 族化合物半導体層をエピタキシャル成長させた後、その後に成長させる III-V 族化合物半導体層の成長温度よりも高い温度での熱処理工程を施しておくので、前記 V 族原子を過剰に含む III-V 族化合物半導体層中に前記 V 族原子のより一層大きい析出物を形成することができ、この結果、転位の伝播をより一層防止することができ、前記 III-V 族化合物半導体層表面近傍における転位密度がより一層低減された化合物半導体基板を製造することができる。

### 【図面の簡単な説明】

【図 1】本発明に係る化合物半導体基板の実施例 1 を模式的に示した断面図である。

【図 2】実施例 1 及び比較例 1～3 に係る化合物半導体基板の GaAs 層表面における転位密度の測定結果を示したグラフである。

【図 3】キャリア濃度と化合物半導体基板表面からの距離との関係を示した曲線図であり、(a) は実施例 1 のものの場合、(b) は比較例 3 のものの場合を示している。

【図 4】転位密度と実施例 2～7 及び比較例に係る化合物半導体基板の含有する過剰の As 量との関係を示した曲線図である。

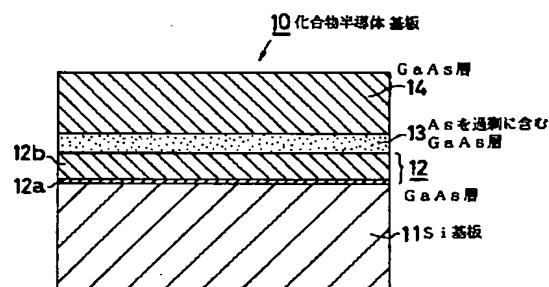
【図 5】実施例 8 に係る化合物半導体基板のヒートバ

ーンを示した図である。

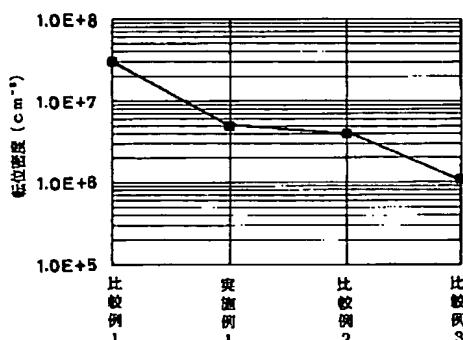
【図6】S i 基板上にG a A s 層がエピタキシャル成長させられ、このG a A s 層中にA s を過剰に含むG a A s 層を1層含んでいる化合物半導体基板における転位の伝播を説明するために示した模式的断面図である。

【符号の説明】

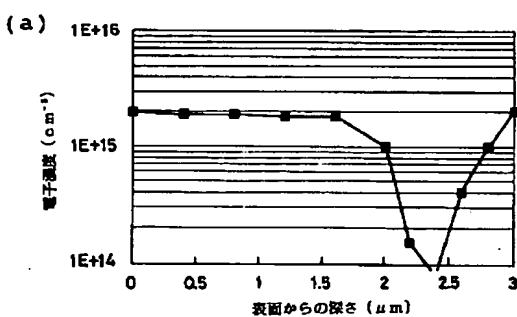
【図1】



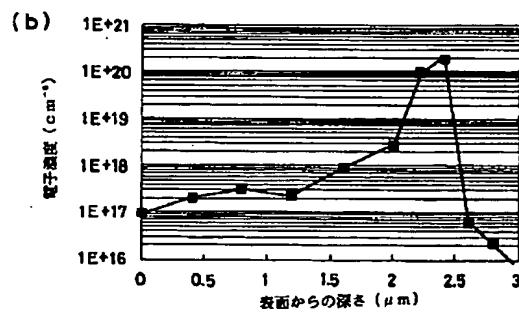
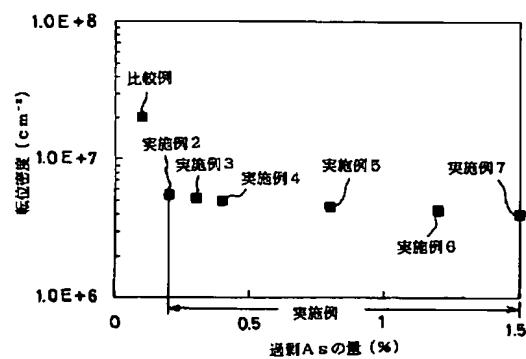
【図2】



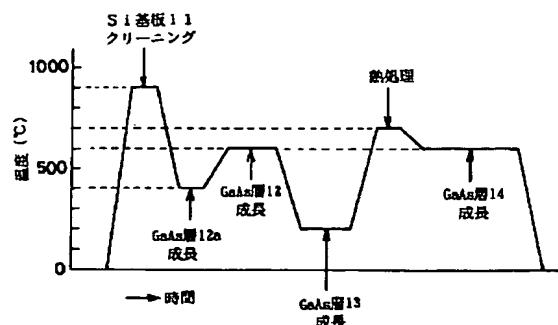
【図3】



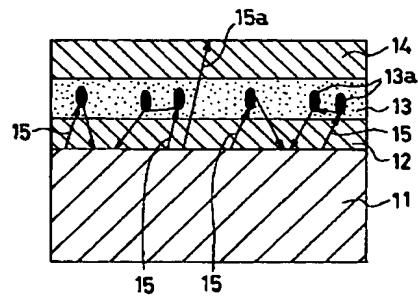
【図4】



【図5】



【図 6】



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